# Combinatorial thermal platform for study of polythiophene thin film structure and OFETs

Leah A. Lucas

Materials Science and Engineering & Flexible Display Center, Arizona State University

ACS Organic Thin Films for Photonic Applications 13 September 2006 3:20PM











Prof. Ghassan E. Jabbour jabbour@asu.edu









Dean DeLongchamp dean.delongchamp@nist.gov



Flexible Display Center ARIZONA STATE UNIVERSITY

http://flexdisplay.asu.edu



**Leah Lucas** 



Joe Kline



Eric Lin



**Brandon Vogel** 



Iain McCulloch Martin Heeney

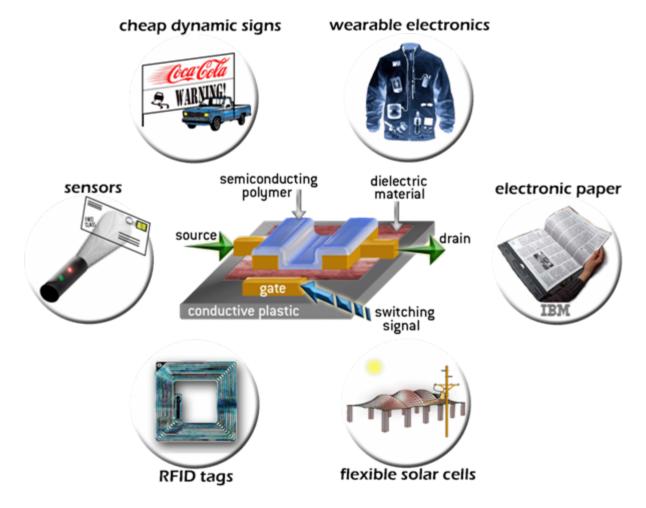
Surface and Microanalysis, CSTL Lee Richter

Ceramics Division, MSEL (NSLS Station) Dan Fischer



Mike Fasolka

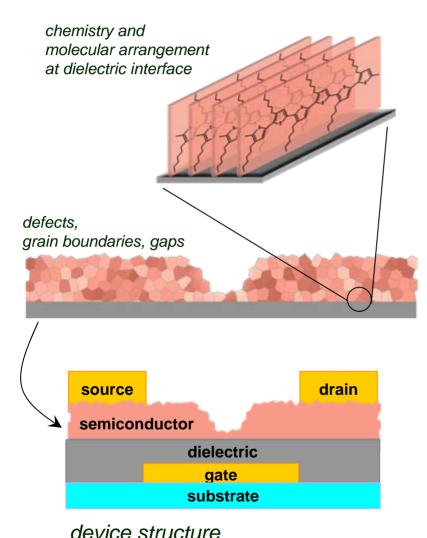
### **Organic Electronics**



#### Alternative technology to Si-based

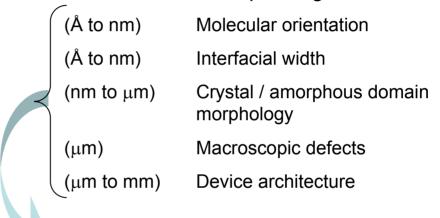
- Lower performance, disposable applications
- Potential advantage → cheaper processing costs
- Solution, large area & R2R processing (flexible substrates)
- New form functionality → conformable, rollable, lightweight

### The dominant role of structure



... and on to circuits

- Structural knowledge required to understand & control electronic properties
- Data needed over multiple length scales



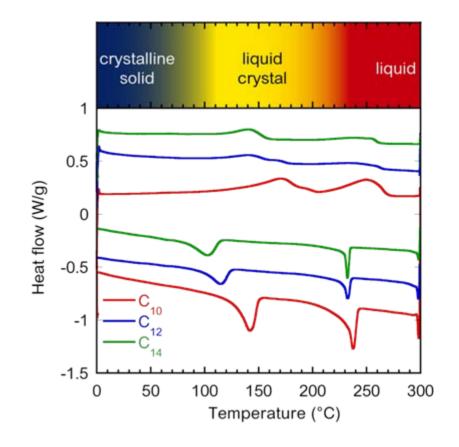
Compromise carrier mobility, device performance, and reliability.

Our goal: correlate processing to structure to performance, with quantifiable structural measurements.

How can we assess structure of organic semiconductors **especially at the buried interface**?

# Truly high performance polymer semiconductors

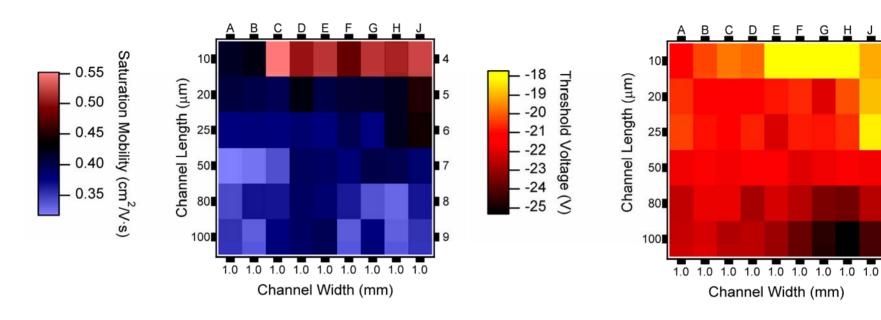
poly(2,5-bis(3-alkylthiophen-2-yl) thieno[3,2-b]thiophenes) (**pBTTT**)



- Solution processible
- Saturation hole mobility 0.2 to 0.6 cm²/V-s (comparable to a-Si)
- Improved mobility after annealing to liquid crystal

	M <sub>N</sub> /M <sub>W</sub> (Da)
pBTTT-C <sub>10</sub>	27,400 / 62,700
pBTTT-C <sub>12</sub>	29,600 / 54,000
pBTTT-C <sub>14</sub>	28,000 / 61,000
pBTTT-C <sub>18</sub>	26,100 / 41,800
<i>DP</i> ≈ 40 - 43	

### **Typical Saturation Mobility & Vt**



- No exotic dielectric (octyltrichlorosilane on oxide)
- No exotic post-spinning treatments (other than annealing)
- Simple processing Just spin and anneal
- In agreement with Merck measurements

What is behind the high performance of the pBTTT? How can we quantitatively assess its structure?

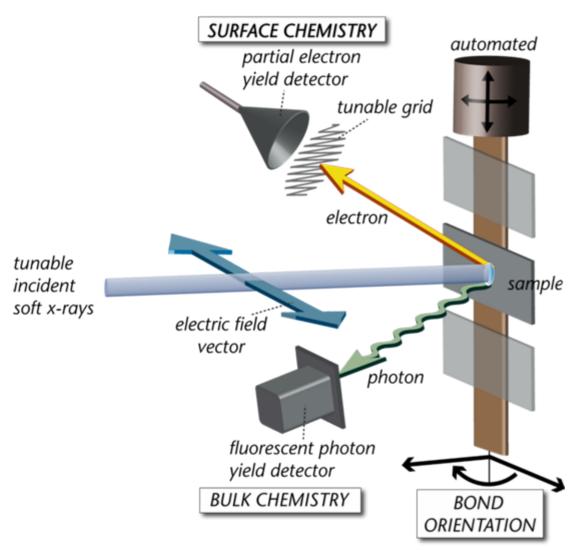
 $\mu_{AVG}$  (cm<sup>2</sup>/V·s)

8

9

	171101
pBTTT-C <sub>10</sub>	~0.2
pBTTT-C <sub>12</sub>	~0.3
pBTTT-C <sub>14</sub>	~0.4
pBTTT-C <sub>18</sub>	~0.2

### **NEXAFS** for structure and chemistry



Near-Edge X-ray Absorption Fine Structure (NEXAFS) Spectroscopy

#### **Strengths for Organic Electronics:**

- Detects C, N, O, & F bonds.
   High sensitivity to π bonding.
- Directly measures molecular orientation.
- · Depth sensitive.
- Collects chemistry and orientation information simultaneously.
- Detects defects.

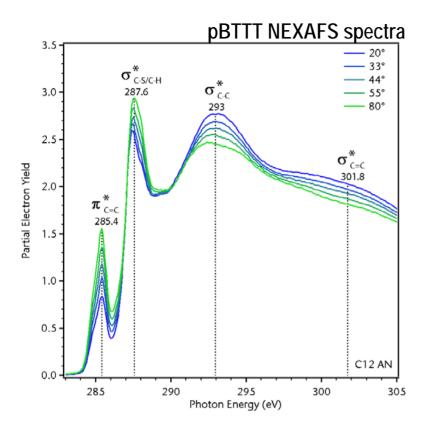
#### **Does NOT measure:**

- Crystal packing style, prevalence, size, shape.
- HOMO, bandgaps.
- Secondary chemical interactions (e.g. vib / rot structure).

A powerful technique complementary to AFM, GIXD, and *XPS*.

Often sufficient for solo investigations.

### **NEXAFS - Conjugated plane tilt**



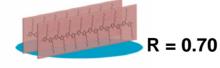
Resonance Intensity:

$$I(\Theta) = \int_{peak} PEY(\Theta)$$

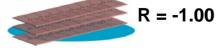
Dichroic Ratio:  

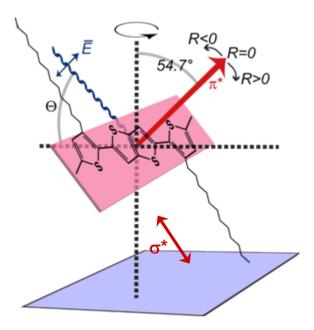
$$R = \frac{I(90^{\circ}) - I(0^{\circ})}{I(90^{\circ}) - I(0^{\circ})}$$

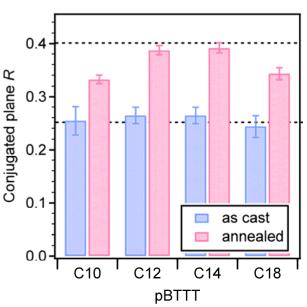
Edge-On



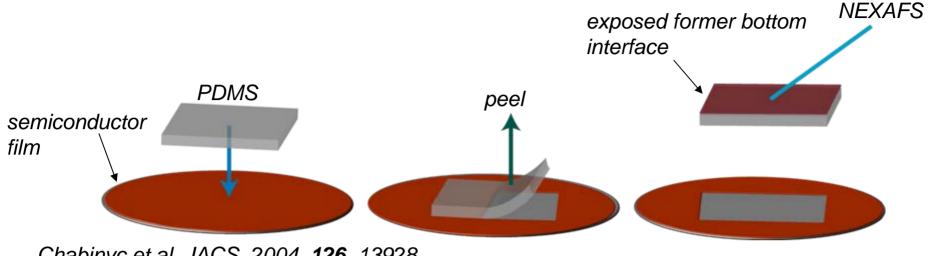
Plane-On



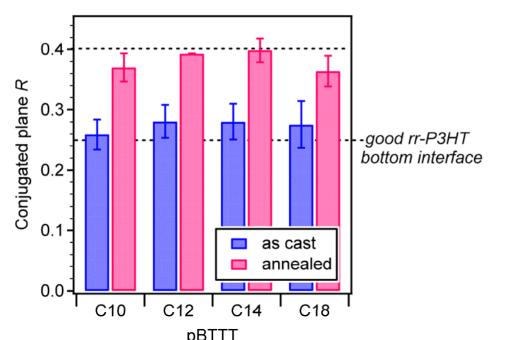




### Ex-situ bottom interface orientation measurement w/ NEXAFS

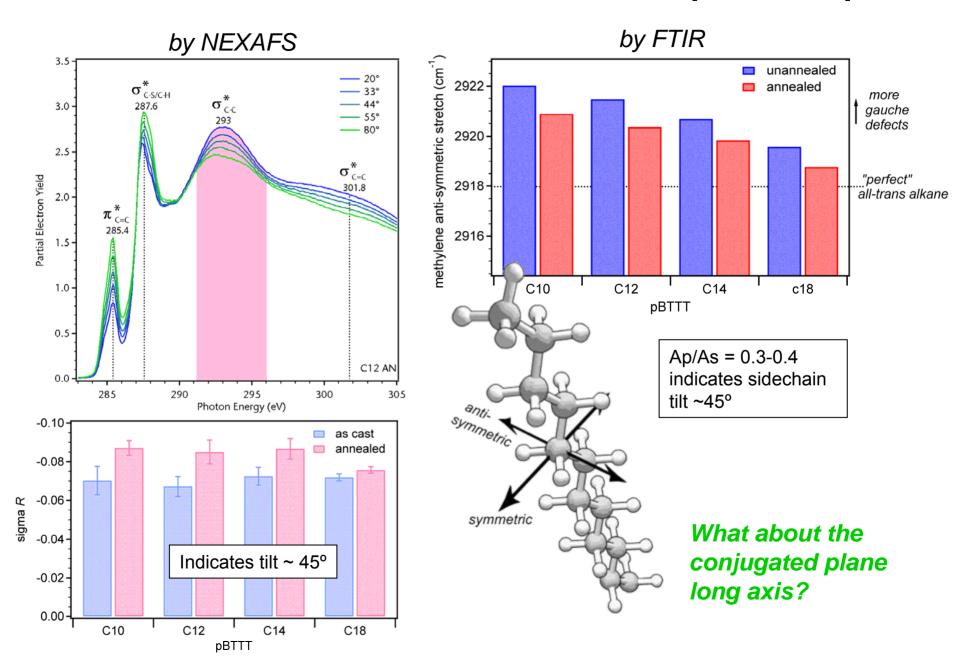


Chabinyc et al. JACS, 2004, 126, 13928

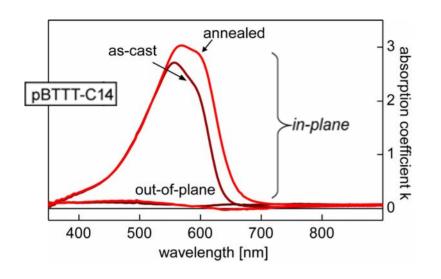


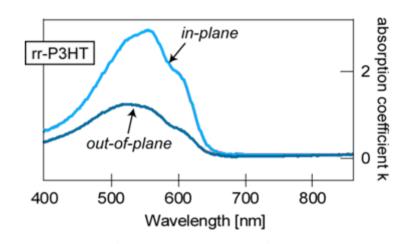
- Bottom interface orientation is very similar to top interface orientation. (Remember films are ~20 nm thick)
- pBTTTs have more vertical conjugated plane than P3HT
- But, the conjugated plane orientation is *not* as vertical as for pentacene / oligothiophenes (R=0.5-0.6)

### Consistent side chain orientation from two spectroscopies

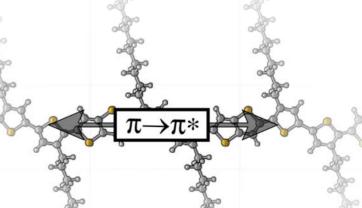


### Polymer long axis orientation by Spectral Ellipsometry

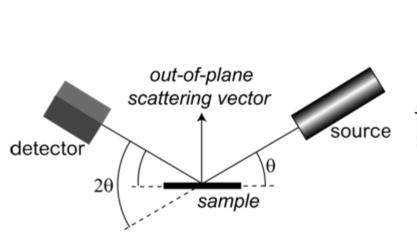




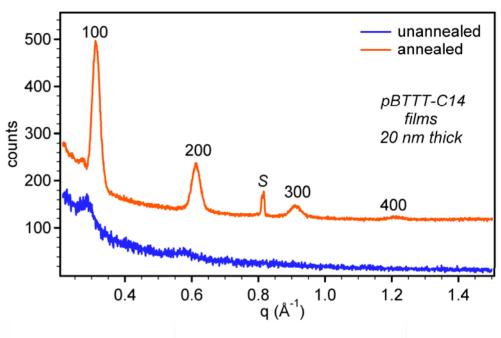
- The  $\pi \rightarrow \pi *$  excitation is oriented along the conjugated long axis.
- pBTTT films exhibit **substantial** long-axis orientation within the substrate plane.

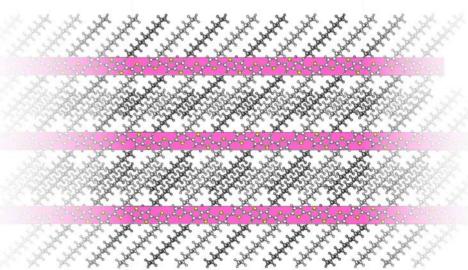


### Specular X-ray diffraction reveals molecular layers

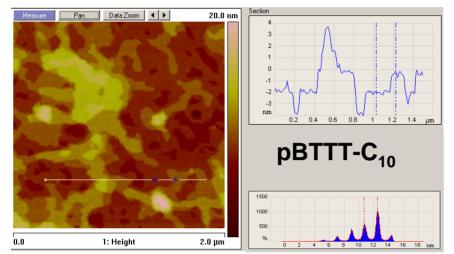


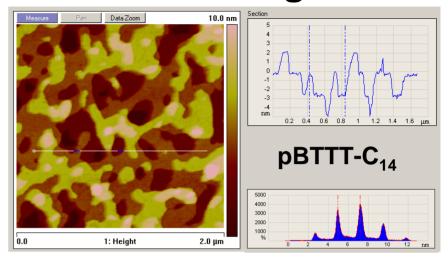
- Clear indication of regularly spaced layers with different electron density.
- Lamellae of packed cores separated from lamellae of packed side chains.
   Also consistent w/ long axis orientation.
- d-spacing ~2.1 nm
- 4 orders of diffraction from 20 nm thick films - indicates highly ordered and oriented structure!

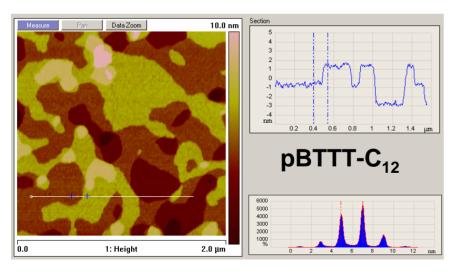


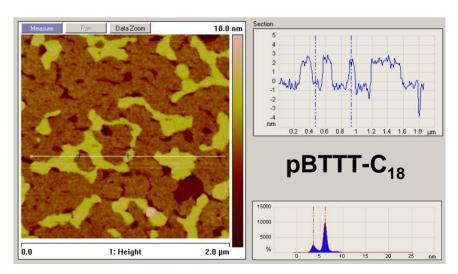


# Atomic force micrographs of annealed films reveal wide terraces of molecular height





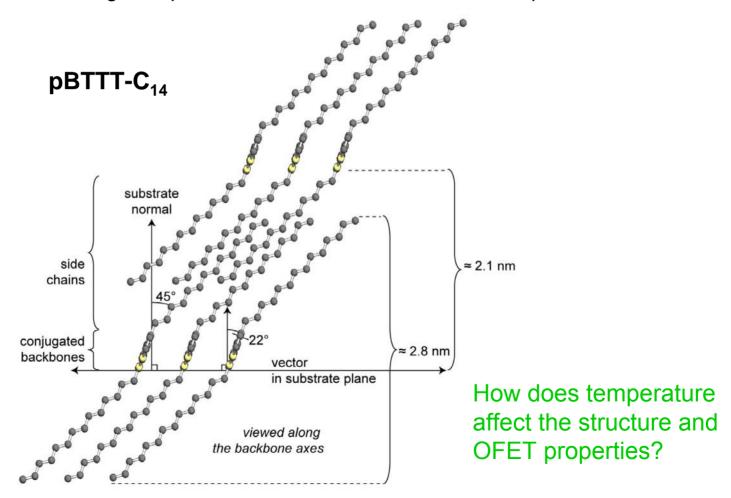




We now know the plane tilt, side chain orientation, the long axis tilt, lamellar packing w/ spacing, and lateral distribution of terraced domains.

### Bringing it together: detailed microstructure of pBTTT-C14

- AFM and X-ray diffraction indicate comprehensive crystallinity of a single type.
- We can assume a narrow, monomodal orientation distribution and extract tilts from spectroscopies
- Further details: DeLongchamp, D. M. et al., Advanced Materials, in press, 2007



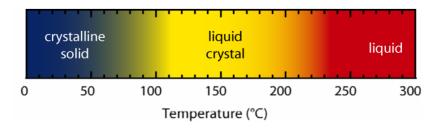
### Combinatorial approach:

### How does temperature affect the structure and OFET properties?

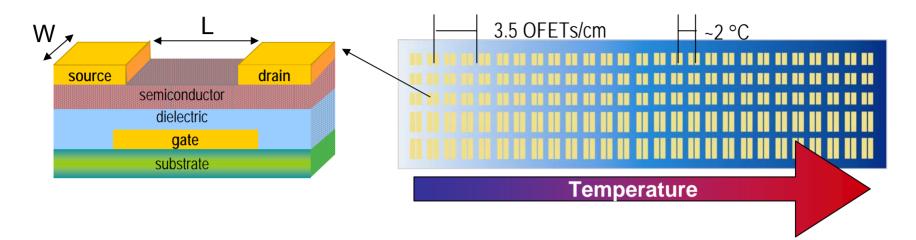


- Distinct structure/morphology difference in as-cast vs. annealed pBTTT films
- Previous structure study used arbitrary annealing temperature
- Combinatorial methods provide means to access this wide parameter space

## Apply a temperature gradient to systematically evaluate the influence of thermal history on pBTTT-C12 OFETs



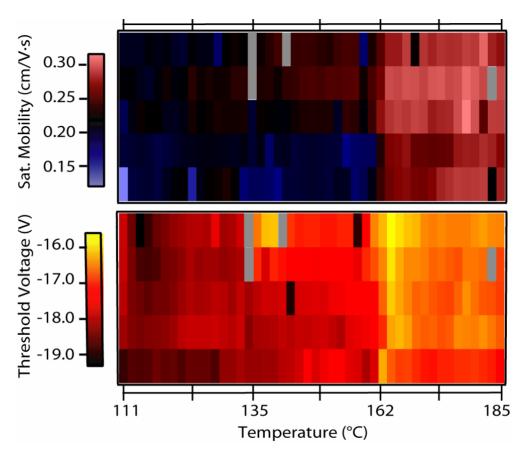
### pBTTT-C12 OFET fabrication & thermal gradient



- pBTTT-C<sub>12</sub> spin coated from warm dichlorobenzene onto OTS-SiO<sub>2</sub>
- Top contact gold S/D electrodes
  - Channel width (W) = 1 or 2 mm
  - Channel length (L) = 80 to 220  $\mu$ m
- Thermal gradient anneal in nitrogen
  - $-\Delta T \sim 75^{\circ}C$
  - Heated for 5 min, cooled to 80 °C (15 min)
- OFET characterization in N<sub>2</sub> purge

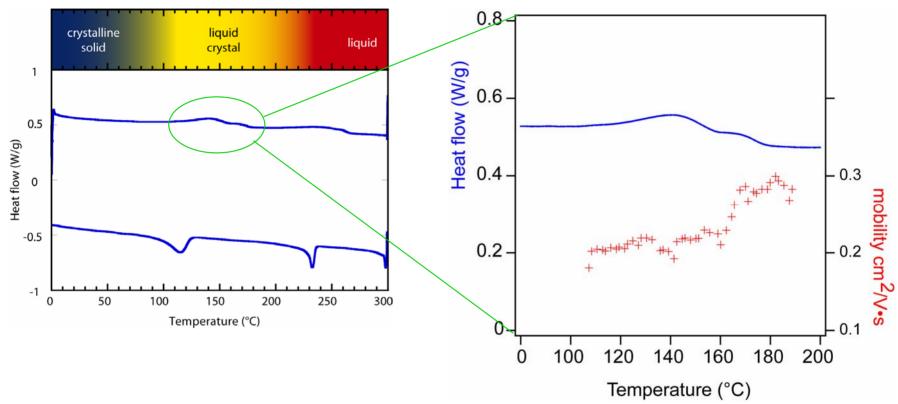
$$I_D = \frac{WC_i}{2L} \mu (V_G - V_T)^2$$

### **OFET characterization of pBTTT-C12**



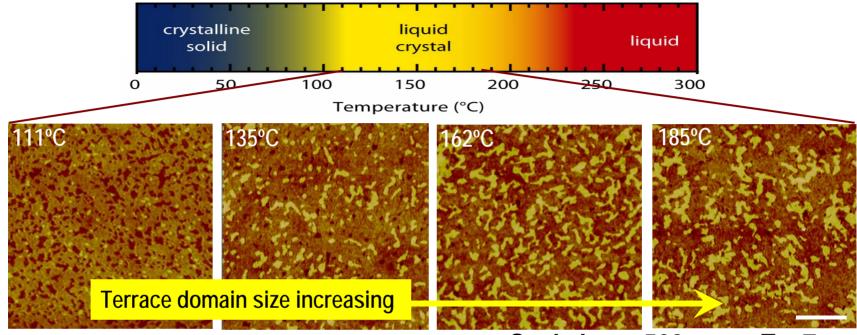
- Each pixel represents individual OFET (230 OFETs processed simultaneously)
- Gray pixels represent OFET with excessive gate leakage
- Mobility increases with temperature with transition region (160 to 165)°C
- Highest mobility ~ 0.3 cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup>

### Differential Scanning Calorimetry of pBTTT-C12



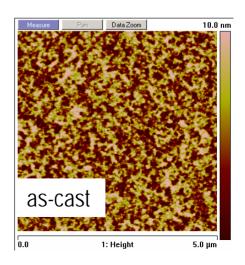
- Increased  $\mu$  coincides with heating into LC phase (DSC)
- Powder DSC reveals melting transition from crystalline solid to liquid crystal occurs at 150-160°C

### **Atomic Force Microscopy of pBTTT-C12**



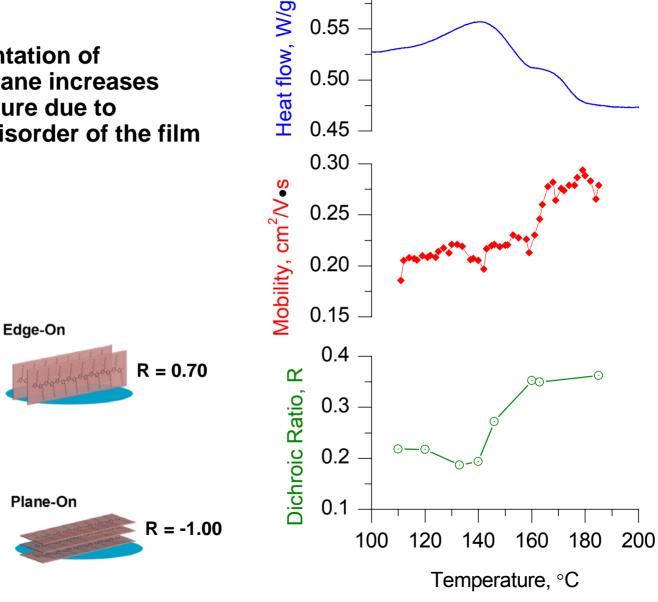
Scale bar = 500 nm,  $\triangle Z = 7$  nm

- As cast-films are featureless
- Lamellar terraces(~2 nm height) form during annealing
- Step height corresponds to lamellar spacing



### **NEXAFS of pBTTT-C12**

**Edge-on orientation of** conjugated plane increases with temperature due to decrease in disorder of the film

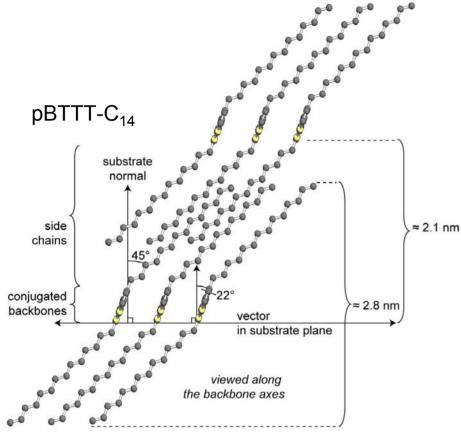


0.60

0.55

0.50

### **Conclusions**



- Improved OFET characteristics result from increased crystalline domain size and molecular ordering when pBTTT-C12 heated into LC state
- Smooth correlation (temperature, structure) made possible by use of combinatorial temperature gradient
- Robust and practical method in new materials development
  - lowest thermal budget necessary
  - screening materials with multiple thermal transitions
  - optimize OFET performance
- Lucas, L. A. et al, Appl. Phys. Lett.
   90, 012112, 2007

### Acknowledgements

\*NIST Combinatorial Methods Center Bell Labs Graduate Research Fellowship Program, Mentor: Alice White